#### LA-UR-13-21205

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Title: R&D for Superconducting RF Structures at Los Alamos

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Intended for: Seminar at Cornell University on 22 Feb. 2013



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### R&D for Superconducting RF **Structures at Los Alamos**

#### Tsuyoshi Tajima

**Mechanical Design & Engineering Group Accelerator Operations & Technology Division** 

Seminar at Cornell University, 22 February 2013 LA-UR-13-21205



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#### **Outline**

- Introduction
- Studies on MgB<sub>2</sub>
- Seamless cavity fabrication (collaboration with KEK)
- New Nb films on copper (collaboration with AASC)
- Photonic band gap (PBG) structure (collaboration with Niowave)





#### Introduction

#### Los Alamos Neutron Science Center (LANSCE)

800-MeV (~1 MW) LANSCE (formerly LAMPF) linac is the heart of the facility -first beam in 1972.

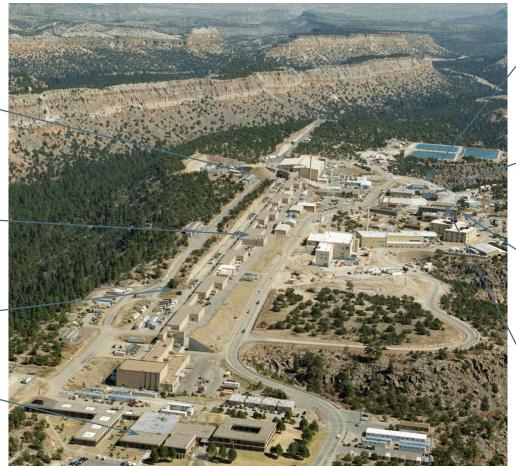
View from west

**Proton Radiography** 

800-MeV Linear Accelerator

Isotope Production Facility

LANSCE Visitor's Center



Manuel Lujan Jr.
Neutron Scattering
Center

Proton Storage Ring



Weapons Neutron Research

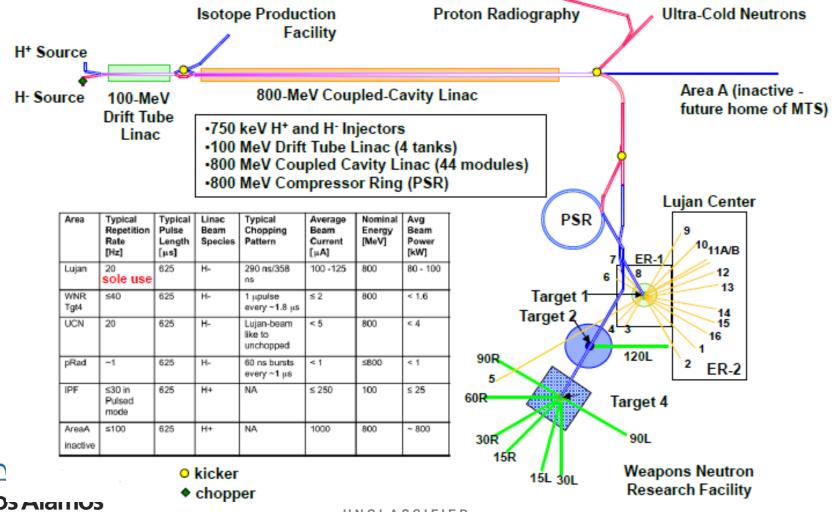
Neutron Resonance Spectroscopy



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## H+ and H- beams are delivered to experimental facilities simultaneously



#### View from east







#### **SRF** Lab facilities

2600 ft<sup>2</sup> Clean Room



Ultra-Pure Water Supply 2000 G/day with 1500 G Tank



Two 38" Diameter Inserts + a 17" insert



140 ft.

100 ft.

38" + 17" Diameter, 10 ft Deep Cryostats and Moveable Shield



**High-Pressure Rinse Equipment** 

**Building MPF-17** 

**Control, Tuning Room** 



BCP, coating and materials science facilities are at different sites in LANL.

## Current R&D is focused on new materials and new fabrication techniques that will enhance performance and reduce cost of future SRF structures

- Recent SRF cavities (especially 1.3 GHz elliptical cells) made of bulk Nb are approaching its theoretical limit of  $E_{acc} \sim 50$  MV/m ( $B_{peak} \sim 200$  mT).
- While it is important to increase the production reliability and yield of high-quality Nb SRF cavities, finding a new material that could overcome the limit of Nb technology is becoming increasingly important for SRF technology to be more attractive and open up other opportunities.

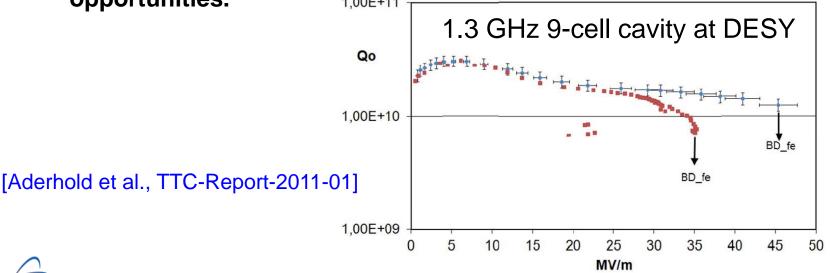




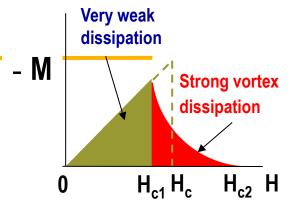
Figure 2: First and final  $Q_0(E_{acc})$  performance of nine-cell cavity AC155 at 2 K after EP treatment. Notice the Q-switches at around 30 MV/m down to 18 MV/m – 24 MV/m at  $Q_0$  below  $10^{10}$ 

### Studies on MgB<sub>2</sub>





#### Type II superconductor



Material	T <sub>c</sub> (K)	H <sub>c</sub> (0) [T]	H <sub>c1</sub> (0) [T]	H <sub>c2</sub> (0) [T]	λ(0) [nm]
Pb	7.2	80.0	na	na	48
Nb	9.2	0.2	0.17	0.4	40
Nb <sub>3</sub> Sn	18	0.54	0.05	30	85
NbN	16.2	0.23	0.02	15	200
MgB <sub>2</sub>	40	>0.43	>0.03	3.5	140
YBCO	93	1.4	0.01	100	150

- Very weak dissipation
- -at H <  $H_{c1}$  (Q =  $10^{10}$ - $10^{11}$ )
- -Q drop due to vortex
- dissipation at H > H<sub>c1</sub>

Nb has the highest lower critical field H<sub>c1</sub>

$$H_{c1} = \frac{\phi_0}{4\pi\lambda^2} \left( \ln \frac{\lambda}{\xi} + 0.5 \right)$$

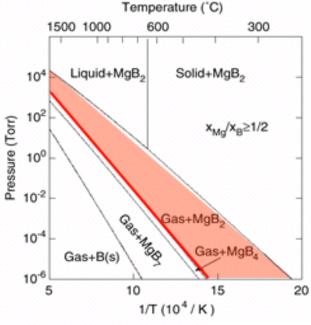
•Thermodynamic critical field H<sub>c</sub> (surface barrier for vortices disappears)

$$H_c = \frac{\phi_0}{2\sqrt{2}\pi\lambda\xi}$$

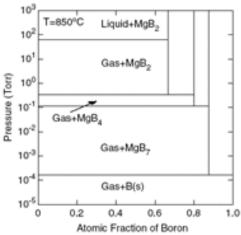




### Keys to Growth of MgB<sub>2</sub> Films



Liu et al., APL 78, 3678 (2001)



#### Keep a high Mg pressure for phase stability

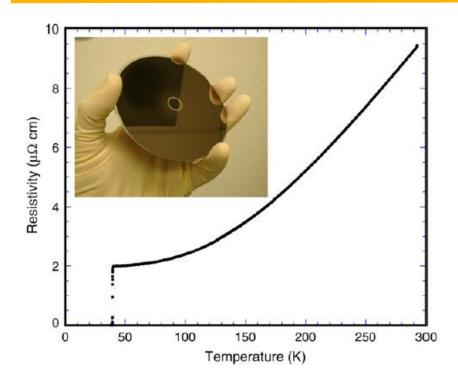
For example, at 600° C Mg vapor pressure of 0.9 mTorr or Ma flux of 500 As is needed

- No need for composition control, as long as the Mg:B ratio is above 1:2.
- Keep oxygen away: Mg reacts strongly with oxygen - forms MgO. reduces Mg vapor pressure.
- Avoid carbon: Carbon doping reduces T<sub>c</sub> and increases resistivity



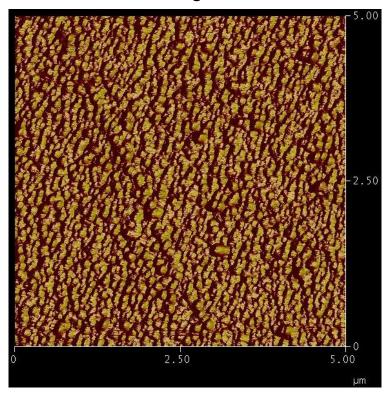
[Xi, Thin film workshop, JLAB, 18-20 July 2012]

## Very smooth films prepared with reactive co-evaporation at STI [Moeckly and Ruby, SUST 19 (2006) L21]



**Figure 2.** Resistivity versus temperature for an  $MgB_2$  film deposited on polycrystalline alumina. This film was grown at 550 °C to a thickness of 550 nm and has a zero-resistance  $T_c$  value of 39.1 K. The inset shows a photograph of an  $MgB_2$  film grown on a 4 inch diameter r-plane sapphire substrate.

500-nm MgB<sub>2</sub> film on Nb RMS surface roughness = 3.0 nm



[D. Oates, SRF Thin Film Workshop 2012]



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# In 2005, we showed that $R_s$ can be lower than that of Nb and there is little increase in $R_s$ with fields up to 120 Oe limited by available power [Tajima et al. PAC2005]

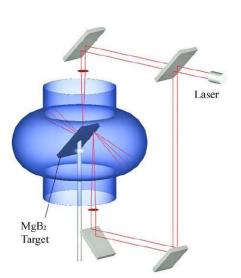


Figure 7: An idea for coating a cavity using a MgB<sub>2</sub> target and a KrF excimer laser.

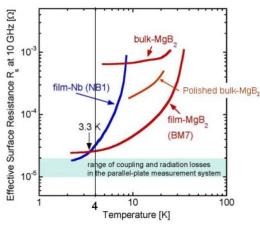


Figure 3: Surface resistance vs. temperature of a 400 nm MgB<sub>2</sub> film coated on a sapphire substrate. Bulk samples and Nb data are shown for comparison. [6]

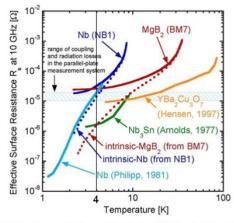


Figure 4: Prediction of intrinsic (BCS) surface resistance (dotted line) from experimental data. [6]

Deposited with reactive co-evaporation at Superconductor Technologies, Inc. (STI)



Figure 5: MgB<sub>2</sub> coated Nb disks of 14.6 mm in diameter.

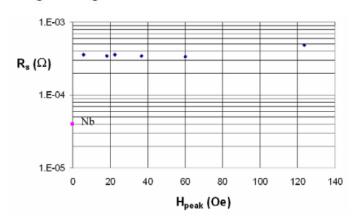


Figure 6:  $R_s$  of the Nb  $TE_{011}$  mode cavity with a  $MgB_2$  sample at the center of the bottom plate, as a function of the peak magnetic field on the sample. The data was converted to 10 GHz using an  $f^2$  law.

Collaboration with Romanenko at Cornell



# In 2005-2006, we proposed to coat cavity with pulsed laser deposition (PLD), but the PLD films had poor quality [Tajima et al. EPAC2006]

#### $T_c \sim 27 \text{ K}$

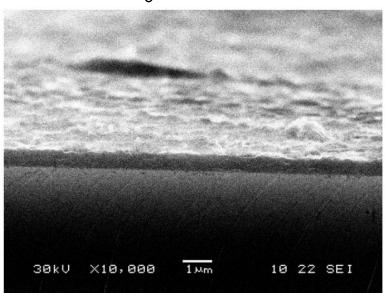


Figure 3: A SEM image of the cross section of off-axis PLD MgB<sub>2</sub> film (ID: 300705v) on Al<sub>2</sub>O<sub>3</sub>-C substrate. The film thickness is 500-700nm

#### Tested by Romanenko at Cornell

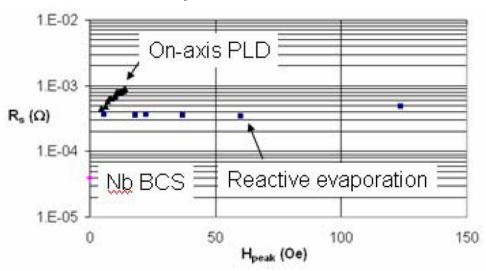


Figure 6: Surface resistance at 10 GHz as a function of surface magnetic field. The data was scaled from 6 GHz data using  $f^2$  law.

#### Collaboration with Yue Zhao of U. Wollongong, Australia



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### Summary of MgB<sub>2</sub> deposition techniques that we have tested

#### Reactive Co-evaporation by Brian Moeckly et al., at STI

Heater pocket

Magnesium

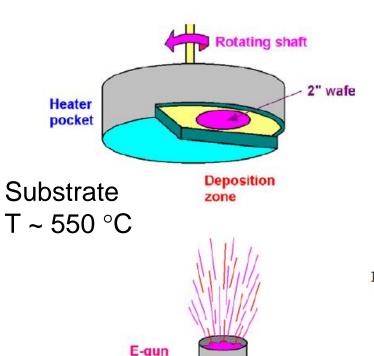


Figure 2: Cross section of the deposition chamber. [5]

[Moeckly et al., ASC2004] [Tajima et al., PAC2005]

Rotating shaft

**Deposition zone** 

Figure 1: MgB2 coating system at STI. [5]

B source

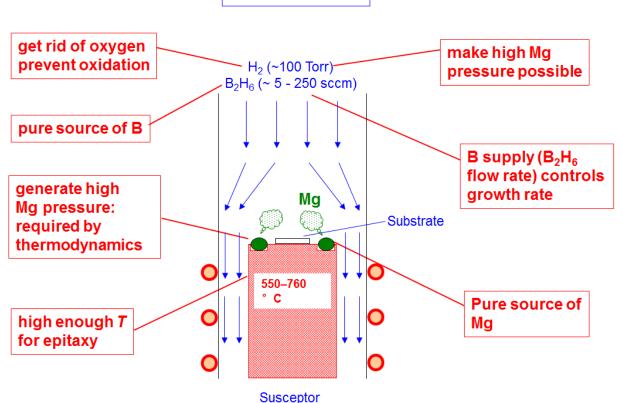
Slide 14

 $T_{\rm C} \sim 39 \text{ K}$ 

## Summary of MgB<sub>2</sub> deposition techniques that we have tested (cont.)

#### **Hybrid Physical-Chemical Vapor Deposition**

**Schematic View** 



 $T_c \sim 40 \text{ K}$ 

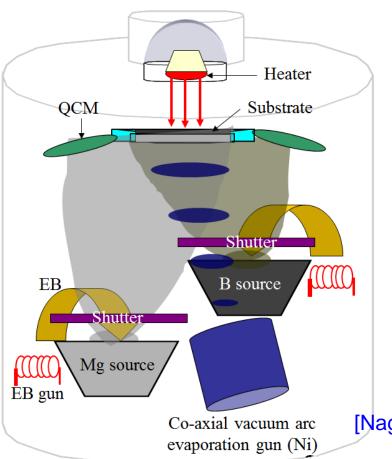
[Xi, Thin film workshop, JLAB, 18-20 July 2012]

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## Summary of MgB<sub>2</sub> deposition techniques that we have tested (cont.)

E-beam co-evaporation by Toshiya Doi et al. of Kagoshima U., Japan



Requires base pressure <~1 x 10<sup>-9</sup>
Torr to avoid effect of oxygen

 $T_{c} \sim 32 \text{ K}$ 

Substrate T ~ 250 °C

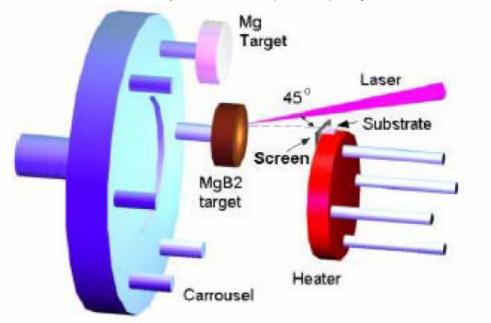
[Nagatomo et al., Physica C 426 (2005) 1459]



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### Summary of MgB<sub>2</sub> deposition techniques that we have tested (cont.)

Off-axis Pulsed Laser Deposition (PLD) by Y. Zhao of U. Wollongong, Australia



 $T_{c} \sim 27 \text{ K}$ 

Substrate T ~ 680 °C

Figure 1: An illustration of off-axis PLD [1].

A KrF laser ( $\lambda$ =248 nm, 25 ns) was used in 120 mTorr Ar atmosphere, then an in-situ annealing was carried out at 680 °C for 2 min in a 760 Torr Ar atmosphere [7].



# In 2007, we started high-power RF tests with 2-inch diameter disks in collaboration with SLAC using 11.4 GHz RF short pulse ( $\sim$ 1 $\mu$ s) and a TE<sub>013</sub>-mode cavity

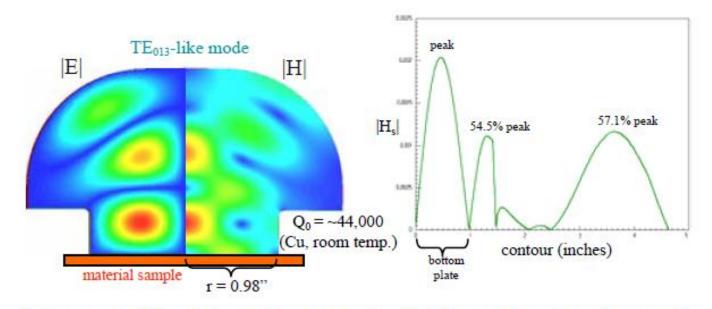


Figure 2: Electric and magnetic fields in the "mushroom" cavity (left) and magnetic field profile along the surface of the cavity (right).

[Tantawi et al. PAC2007]



### Due to the low Q of the host cavity made of copper, R<sub>s</sub> did not have enough sensitivity. Also, thermal effect seems to have been involved despite a short pulse is used.

STI film deposited at 550 °C on top of ALD alumina

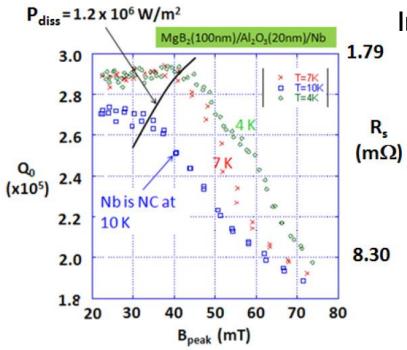


Figure 11: Q<sub>0</sub> as a function of B<sub>peak</sub> at 4 K, 7 K and 10 K for the MgB<sub>2</sub>(100 nm)/Al<sub>2</sub>O<sub>3</sub>(20 nm)/Nb sample shown in Fig. 8. The solid line is a curve described by Eq. (2) with  $P_{diss} = 1.2 \times 10^6 \text{ W/m}^2$ .

[Tajima et al., SRF2011]

Inter-diffusion layer causes high RF loss

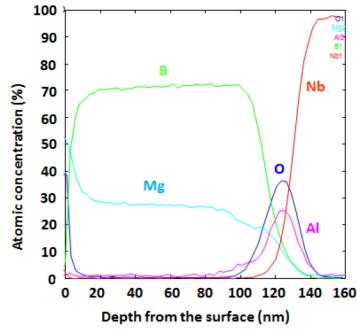


Figure 12: Auger depth profile of the MgB<sub>2</sub>(100 nm)/Al<sub>2</sub>O<sub>3</sub>(20 nm)/Nb sample shown in Fig. 8. Interdiffusion of coating components at the interfaces is observed.

Slide 19

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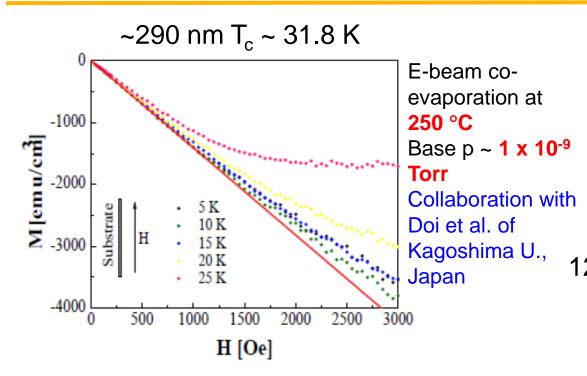


Figure 4: Magnetization curves as a function of applied magnetic field at various temperatures for  $\sim\!290$  nm thick MgB<sub>2</sub> film ( $T_c\sim31.8$  K) deposited on a Si substrate at Kagoshima Univ.

[Tajima, Haberkorn, Civale et al. LINAC2010]

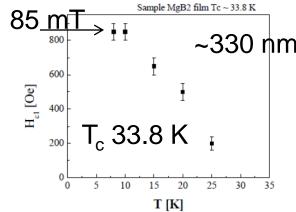


Figure 7:  $H_{c1}$  vs. sample temperature from magnetization measurements of  ${\sim}330$  nm thick MgB<sub>2</sub> sample prepared by Kagoshima Univ.

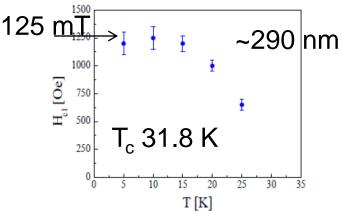


Figure 8:  $H_{c1}$  vs. sample temperature from magnetization measurements of ~290 nm thick MgB<sub>2</sub> sample ( $T_c \sim 31.8$  K) prepared by Kagoshima Univ.

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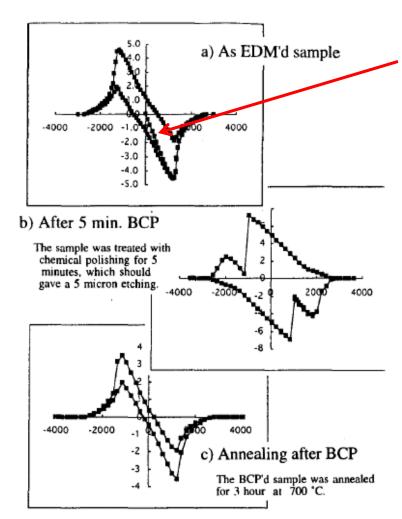


Fig. 2. Hysteresis appeared in magnetization curves. a) as EDM'd sample, b) after chemically polished the EDM'd sample, c) annealed after the CP'd sample.

#### We are looking at only this part

Magnetization measurements were used for SRF cavity research in 1995 by Saito and Wake of KEK to evaluate the effect of surface treatments by looking at hysteresis curves.

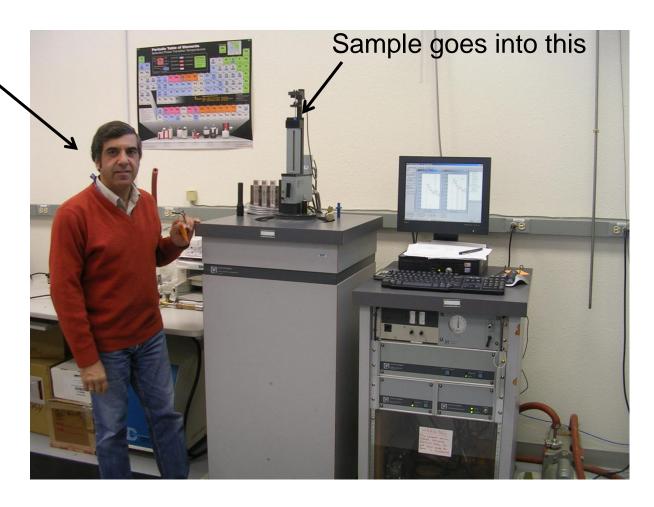
The larger hysteresis means more imperfections and/or dislocations since vortices get pinned by them.

[Saito, Wake, SRF1995]



## Recent studies focused on $B_{vp}$ measurements using a Quantum Design SQUID magnetometer since it seems to be a reliable method to determine fundamental limit

At LANL, Leonardo Civale and his postdocs have been carrying out these measurements





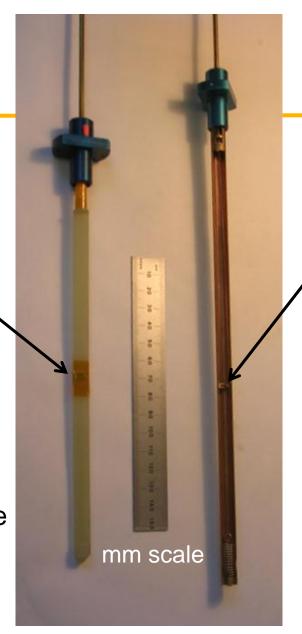


#### **Samples**

Typically ~5 mm x ~5 mm for coated films

Nb reference was a rod with 2 mm diameter and 10 mm long cut out from a single grain RRR>300 Nb sheet

Rod or ellipsoid is better due to less edge effect, but difficult to uniformly coat

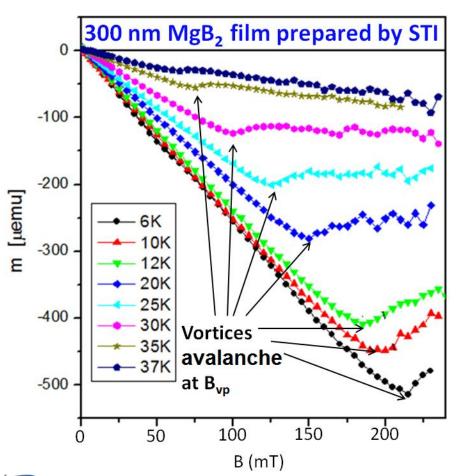


This sample holder has an angle adjustment mechanism

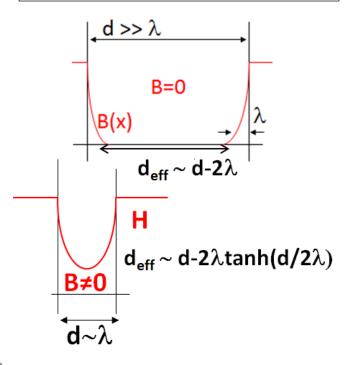
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# Meissner slope is proportional to the effective volume of the film, making it difficult to measure ultra-thin films due to small signal. Our present limit is ~200 nm



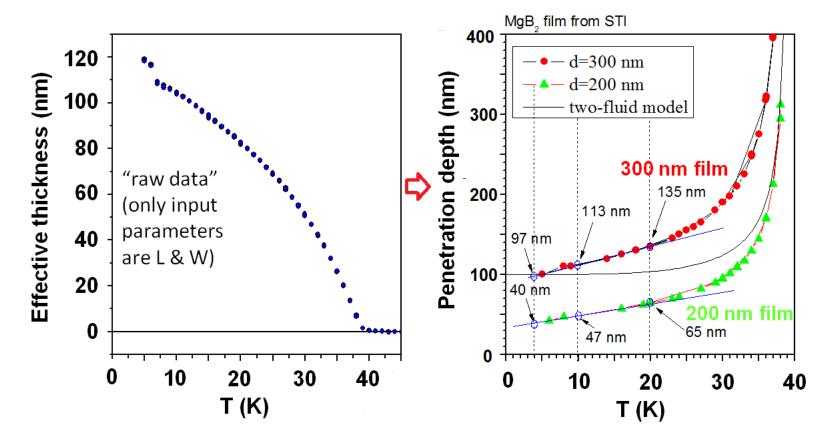
Meissner state: dm/dH =  $-V_{eff}/4\pi \propto d_{eff}$ Slope changes with T due to the change in  $\lambda$ 





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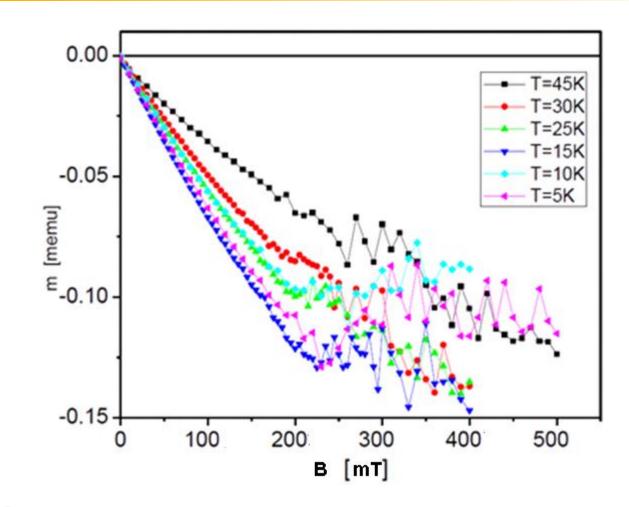
# Penetration depth can be estimated from effective thickness. Penetration depth increases with higher temperature. This causes the reduction of effective thickness.





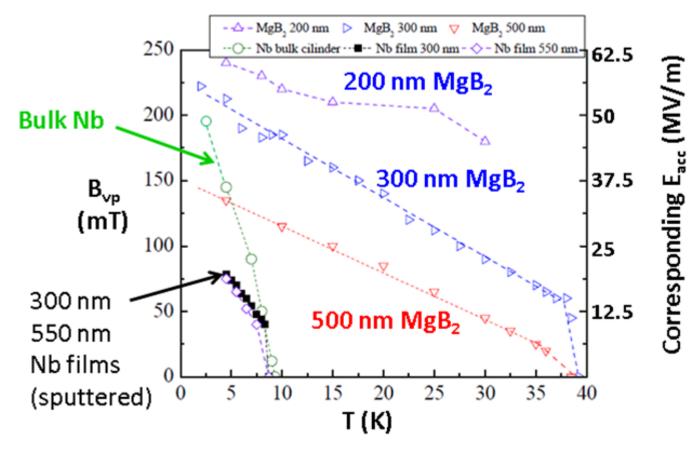


## With a 200 nm sample, the signal got quite noisy at high fields, but was still measurable.





# Summary of $B_{vp}$ for STI films (200, 300 and 500 nm) compared with cavity-grade bulk Nb and sputtered Nb film. MgB<sub>2</sub> thin films show remarkably high $B_{vp}$ !!



Assume  $B_{peak}/E_{acc} = 4 \text{ mT/(MV/m)}$ 

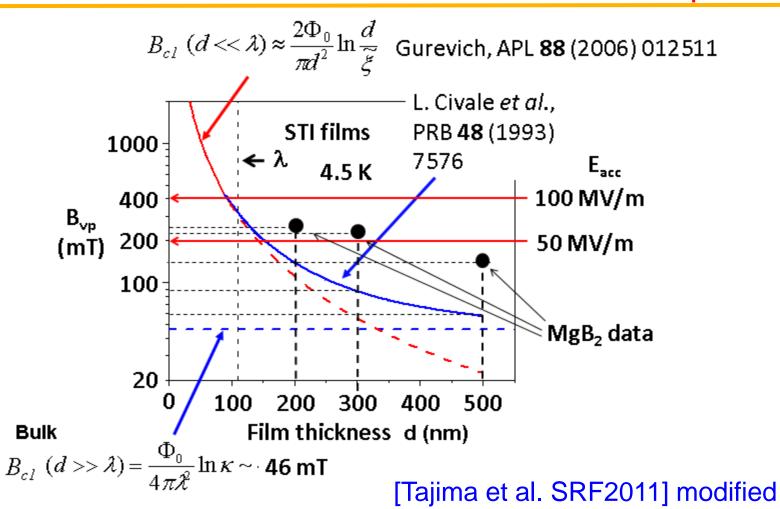




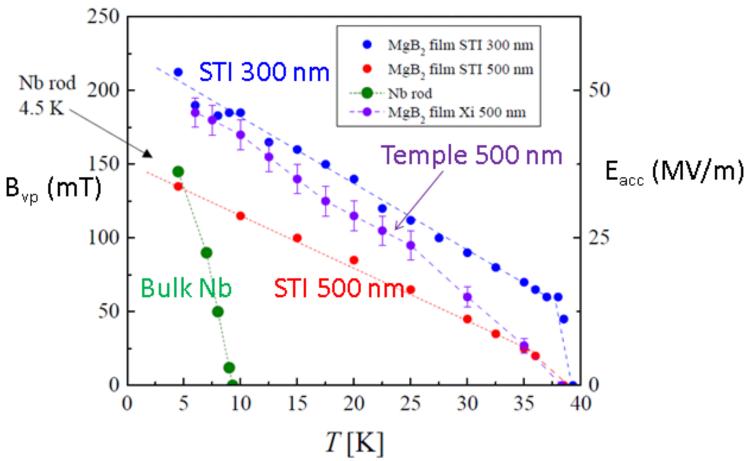




# Comparison with theoretical curve of $B_{c1}$ for thin films assuming $\lambda = 110$ nm and $\xi = 6$ nm. Important finding here was that even the films with d> $\lambda$ have high $B_{vp}$



### Temple University (Xiaoxing Xi's group) samples prepared with hybrid physical chemical vapor deposition (HPCVD) also showed high B<sub>vn</sub>

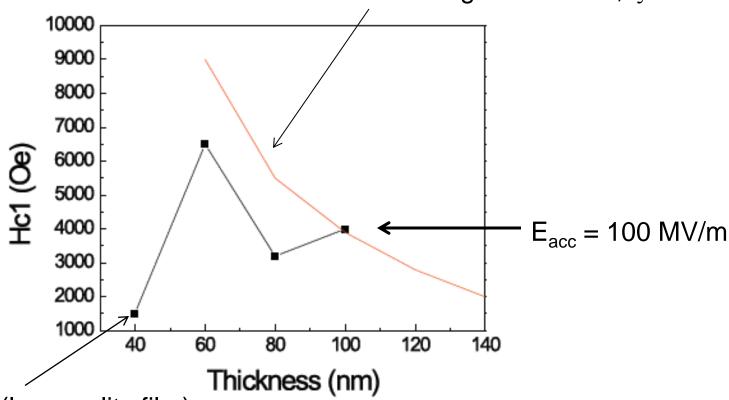




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# Recent results of thinner films prepared by Temple U. and measured by Beringer of the College of William and Mary showed $B_{c1} \ge 400 \text{ mT}$ ( $H_{c1} \ge 4000 \text{ Oe}$ ) with $\le 100 \text{ nm}$ films

Prediction assuming  $\lambda = 140$  nm,  $\xi = 5$  nm



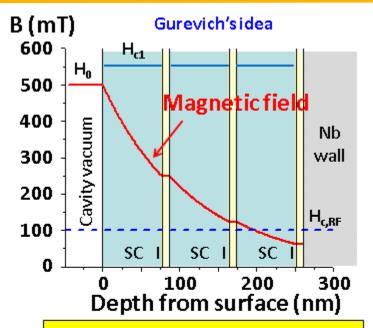
Lower T<sub>c</sub> (low quality film)

[Beringer, Thin film workshop, JLAB, 18-20 July 2012]

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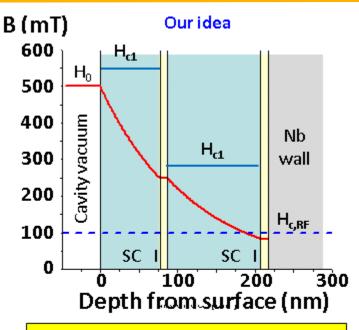


## There is a possibility that we can achieve $\sim$ 500 mT ( $E_{acc}$ $\sim$ 125 MV/m) even with 2 layers of MgB<sub>2</sub>





- $d \le 77$ nm for H<sub>d</sub> ≥ 5500 Oe
- 3 layers needed
- coating curved walls with very thin uniform of layers is challenging



#### Variable thickness multilayers:

- d<sub>1</sub> ≤ 77nm for H<sub>d</sub> ≥ 5500 Oe
- only 2 layers needed
- 2<sup>nd</sup> layer is thicker: 100nm ≤ d<sub>2</sub> ≤ 120nm

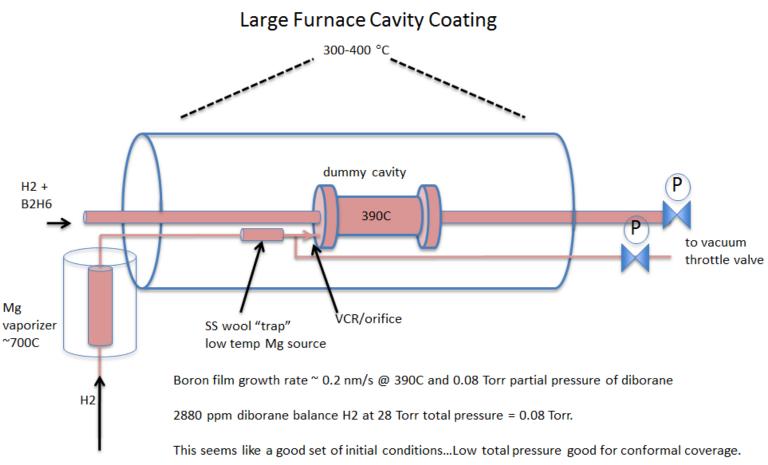


[Tajima et al., SRF2011]

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## At LANL, 2-stage coating, i.e., B layer using diborane (B<sub>2</sub>H<sub>6</sub>), then react it with Mg vapor, is being tried



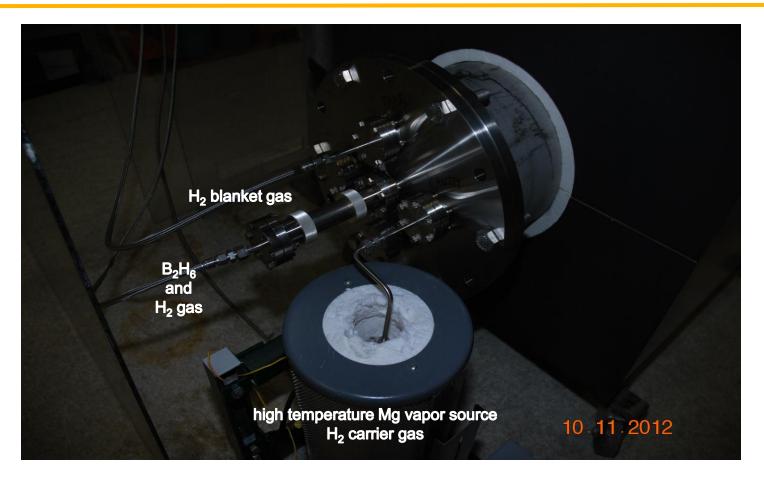


### MgB<sub>2</sub> CVD reactor for a 1.3 GHz cavity located at Technical Area 35





### Gas delivery and external Mg vapor source







### **Exhaust system with excess di-borane burner**







## **Mounting test cavity**





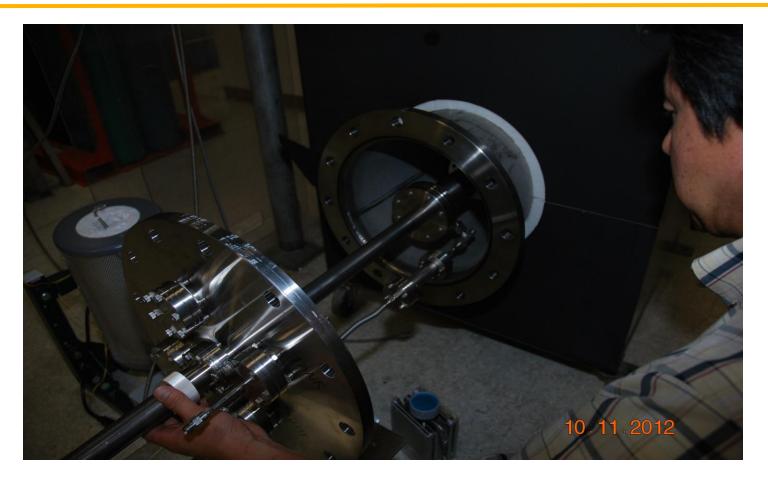
# Copper surrogate cavity with 28 holes for samples. First substrates will be 6 mm x 6 mm x 0.43 mm sapphire







## **Cavity installation into furnace**







### Our plans and milestones for MgB<sub>2</sub> coating

- Achieve superconductivity with witness samples by 30 May 2013
- Optimize coating parameters to get high-quality films by 31 July 2013
- Optimize coating parameters to get good uniformity on cavity surfaces by 30 September 2013
- Coat 1.3 GHz single-cell cavities and test them after that. (Cavities to be coated will be either Nb/Cu or Nb cavities)
- We are currently considering ALD for dielectric coating (Al<sub>2</sub>O<sub>3</sub> or other material)





# Seamless cavity fabrication

# In collaboration with KEK and FNAL I was at KEK as an invited investigator from April 2011 through March 2012





### Status of copper forming at KEK

- Use of C1020 copper seamless tubes (130 mm OD, 3-4 mm-thick) from Hitachi Cable Co., Ltd.
  - Manufacture of copper seamless tubes is an established process in industry.
  - Very inexpensive
    - X50 cheaper than RRR300 Nb sheets.
    - X100 cheaper than appropriate Nb seamless tubes (~\$180 vs. ~ Nb \$17,300 for a 800 mmL, 3 mm-thick tube appropriate for cavity cell forming)
  - Nb films comparable with high-RRR bulk Nb might be available soon
- KEK has been successful in forming 3-cell cavities from these tubes with 500 °C x 2 h vacuum annealing before forming.





# Necking machine at KEK. The length is optimized for an ILC-type 9-cell cavity

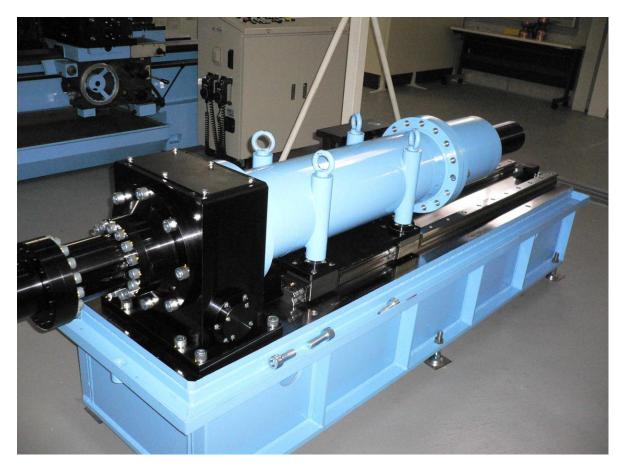






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## Hydroforming machine at KEK. The existing machine cannot form more than 3 cells.





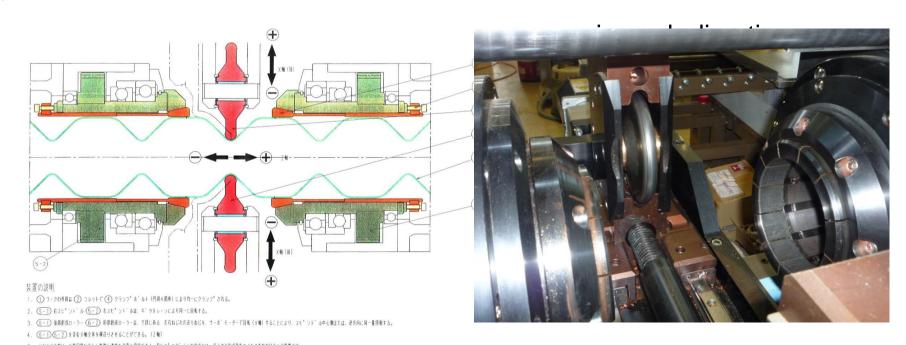
Ueno et al., Kasokuki Gakkai, 2006

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# **Detail of KEK forming process**

### Necking

The pipe rotates at 200 rpm, the moving speed of the roller is 150 mm/min axially and its movement is controlled by NC.





Ueno et al., Kasokuki-Gakkai, 2007

### Some photos from the process to hydroform a copper 3-cell ILC baseline shape



After the first hydroforming using an intermediate constraint dies







Inserted into the cylinder

Slide 45

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## To be attached to the hydroforming machine

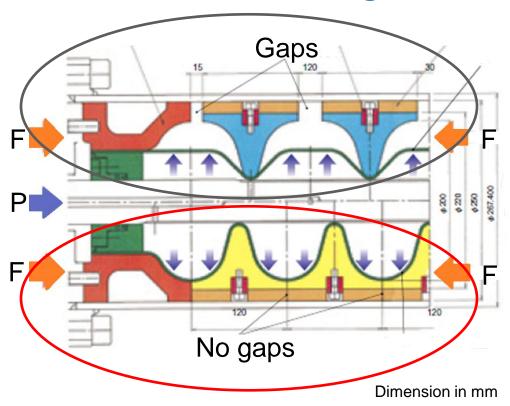




# Hydroforming procedure at KEK (in the case of copper)

- Start with 8 MPa, axial speed is 20-50 mm/min. Try to leave 5-8 mm radial gap between the deformed cell and the die before 2 dies make contact.
- After putting 2 dies together, raise the pressure to 25 MPa for the cells to conform to the dies completely. (From looking at the pressure gauge, it looks like the pipe is being deformed at a pressure 9-13 MPa.)

### **Before forming**



**After forming** 



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## After forming. Conformed to the die very well.





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## After forming a 3-cell cavity





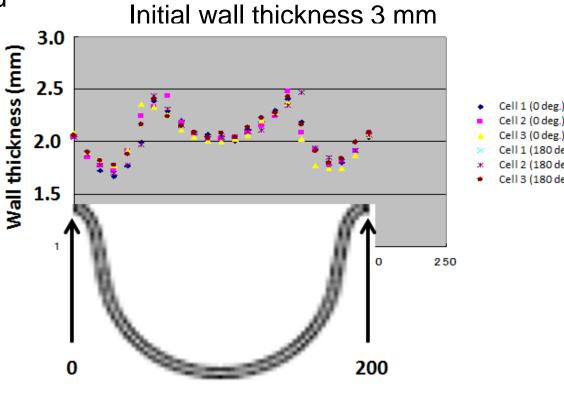


## The starting thickness of 3 mm seems not enough $\Rightarrow$ changing to 3.5 mm and 4 mm tubes

Initial material: Oxygen Free Cu (C1020) seamless pipe from Hitachi Cable Co.



An ILC baseline-cell-shape cavity formed in May 2011 by Hitoshi Inoue



Annealing at 500 °C for 2 hours between necking and hydroforming

Distance along the cell (mm)

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Cell 3 (0 deg.)

Cell 1 (180 deg.) Cell 2 (180 deg.) Cell 3 (180 deg.)

### **Status of Nb forming at KEK**

- Nb is difficult due to much less ductility than copper
- Few vendors that can supply seamless tubes adequate for cavity forming
- FNAL will soon provide KEK with some tubes from ATI Wah Chang, a vendor that was able to produce some formable tubes in the past
- Collaboration with a Japanese vendor is ongoing, but has not been successful in producing an appropriate tube yet.





# New Nb films coated on copper cavity

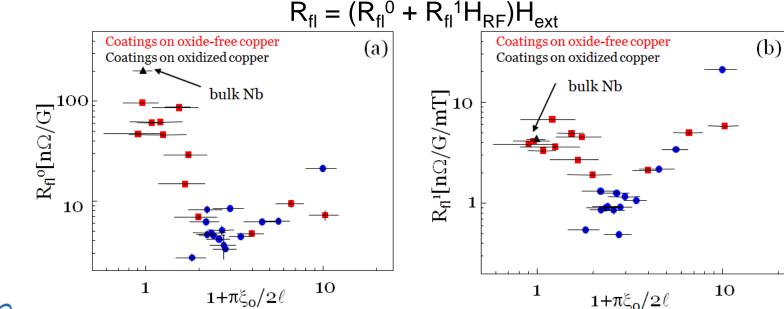
### In collaboration with AASC and KEK





### Advantages of using copper cavity with Nb coating

- Cheaper (x50 less than RRR~300 Nb)
- Due to high thermal conductivity of Cu, surface temperature is lower than bulk Nb ⇒ more stable against quench
- Residual resistance less sensitive to external magnetic field





Calatroni, TTC meeting, JLAB, 07 November 2012.

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# As many of us know, while Nb/Cu cavities were successful for LEP at CERN, sputtered Nb films have strong $Q_0$ slope

### State-of-the-art 25 years ago

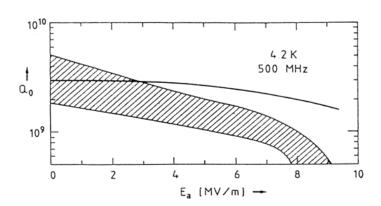
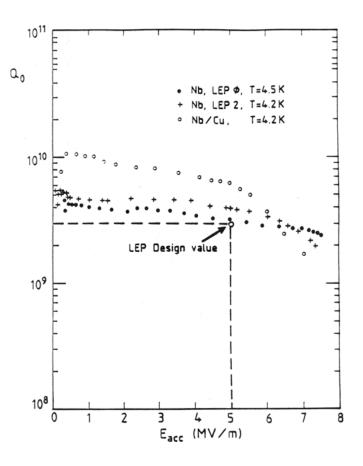


Figure 1. Typical  $Q(E_a)$  curves of sheet metal (line) and sputter coated cavities (hatched).

### [Calatroni, Thin film workshop, Padua, 2006]



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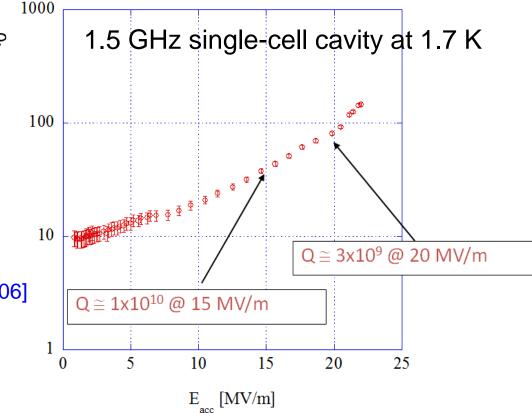


# Parameter optimization led to $\sim$ 28 MV/m, but a strong Q<sub>0</sub> slope still exists

State-of-the-art 5 years ago

The reported highest gradient for Nb/Cu cavities so far is 28 MV/m

[Calatroni, Thin film workshop, Padua, 2006]



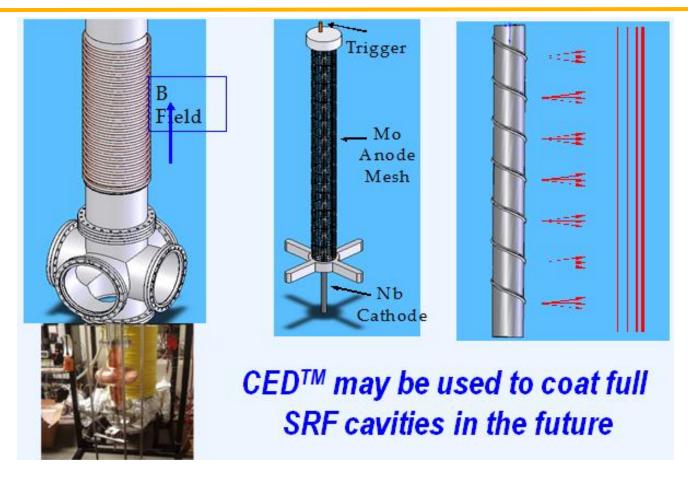


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Rs [nOhm]



## Coaxial Energetic Deposition (CED™) by Alameda **Applied Sciences Corporation (AASC), San Leandro,** California, USA



[E. Valderrama et al., CEC-ICMC2011, Spokane, WA, 13-17 June 2011]



# Coaxial Energetic Deposition (CED™) by AASC [Krishnan et al., PR-STAB 15 (2012) 032001]

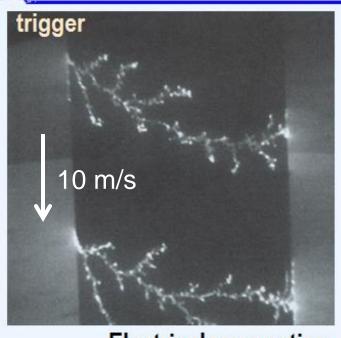
Base pressure after 120 °C bake ~5 x 10<sup>-8</sup> Torr



High RRR films

Needs ~4000 pulses for 1 μm coating

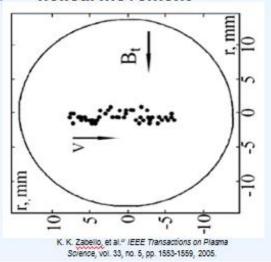
### ROTATING VACUUM ARC PLASMA (CEDTM)



Electrical connection

"Cathodic arcs from fractal spots to energetic condensation". A. Anders, New York: Springer, 2008.

- Retrograde motion pushed the cathodic spot to rotate
- Electrical connection made on the end of cathode
- => helical movement





[E. Valderrama et al., CEC-ICMC2011, Spokane, WA, 13-17 June 2011]

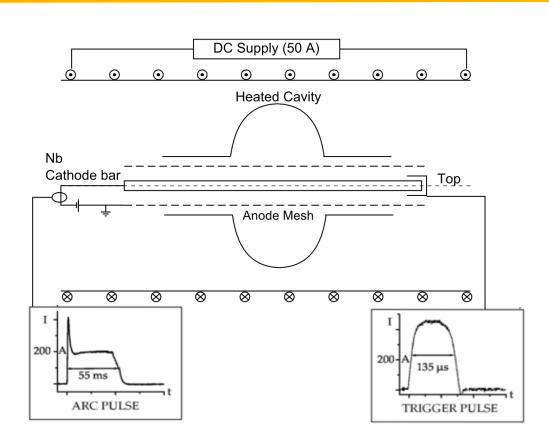
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# Arc source is scalable to high throughputs for large scale cavity coatings. Present version deposits ~ 1 monolayer/pulse ~ 1 ms UHV and clean walls are important



**Coaxial Energetic Deposition (CED™)** 

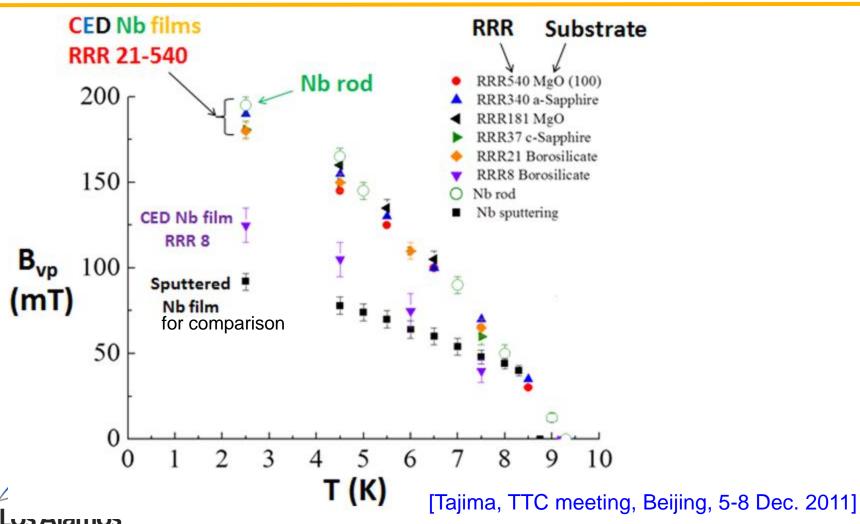






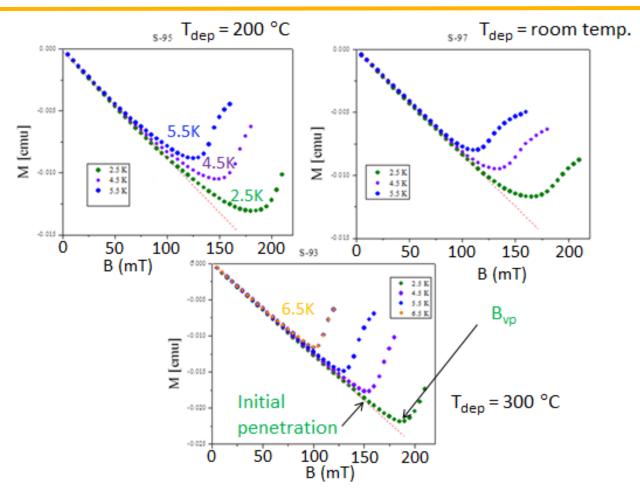
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# B<sub>vp</sub> of CED Nb films coated on dielectric substrates have shown values comparable with RRR~300 Nb





# Magnetization measurements of CED Nb films coated on Cu





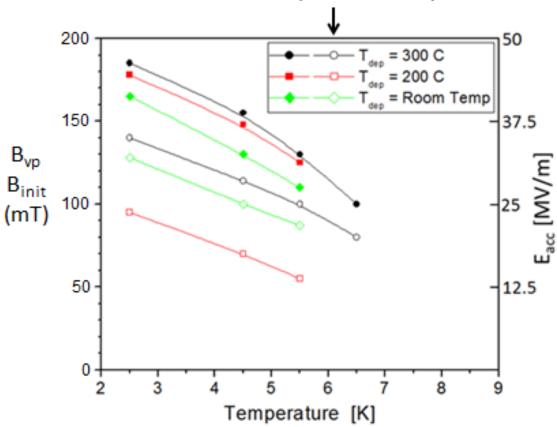
[Civale et al., to be submitted to a journal with more recent data]

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# Summary of magnetization measurement results of CED Nb films coated on Cu

Initial vortex penetration (2 % deviation from Meissner slope)





[Colt James et al., ASC2012, Portland, 7-12 Oct. 2012] modified

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# Coating of 1.3 GHz single-cell copper elliptical cavity with a CED Nb film

3-cell cavity was formed from a seamless copper (C1020) pipe at KEK



Cut into 3 single cells and beam pipes were EB-welded and SST flanges were brazed.



Being used as a surrogate cavity for MgB<sub>2</sub> coating tests



Being used for CED Nb coating tests

# Coating of 1.3 GHz single-cell copper elliptical cavity with a CED Nb film (cont.)

- At LANL, Cu cavity was chemically etched, high-pressure rinsed with ultra-pure water, pumped down, baked at 50-60 °C, filled with filtered N<sub>2</sub> gas and shipped to AASC.
- AASC coated it at 350 °C to a thickness of ~5 μm and shipped back to LANL.
- At LANL, the cavity was HPR'ed at 500 psi for a total of 2 hours in a class 100 clean room, assembled with flanges and couplers and tested. No baking.

[Colt James et al., ASC2012]

After ultra-pure water rinsing at 500 psi for 1 hour in the SRF lab clean room at LANL

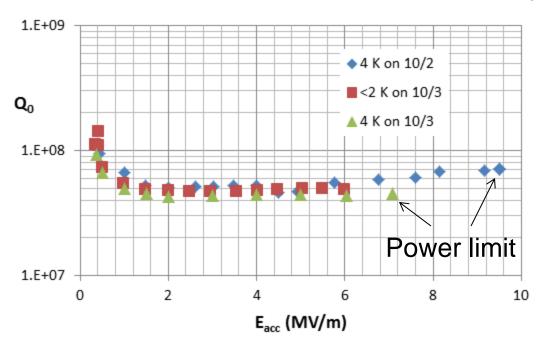




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# The beam pipe (120 mm long) was found to be too short to eliminate the effect of RF losses on the SST end flanges, but it did not quench up to available power limit (~9.5 MV/m)

Q<sub>0</sub> was dominated by the losses on the end flanges.



This result was predicted before the test by MWS calculations and it was verified by this test.



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Pick-up coupler

Input coupler (movable)



### Next steps for Nb/Cu cavity development

- KEK fabricated 3 new hydroformed copper cavities with 170 mm (50 mm longer than existing) beam pipes to eliminate the effect of end SST flanges.
- KEK will send them to LANL by early March 2013.
- These cavities will be coated with Nb at AASC in March 2013.
- They will be tested at LANL in April 2013.
- Some of these cavities will be used for MgB<sub>2</sub> coating if they are good.





# Superconducting photonic band gap (PBG) cavity development

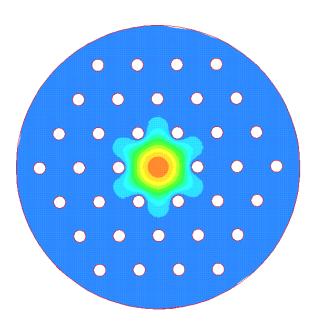
# Led by Evgenya Simakov Funded by DOE SC/HEP and ONR



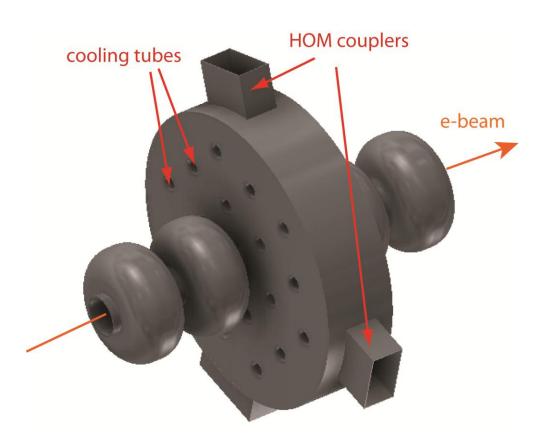


# Effective for HOM damping without sacrificing real estate gradient

### Confines single mode TM<sub>01</sub>



 $Q_{ex, HOM} < 115$ 



Higher  $f \Rightarrow$  lower bunch charge + smaller footprint

Los Alamos NATIONAL LABORATORY

# Fabricated and BCP'ed at Niowave, then assembled and tested at LANL





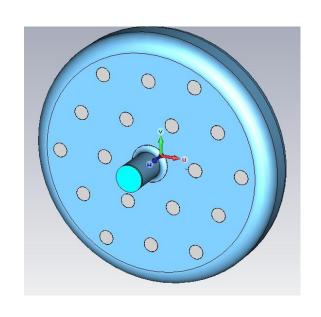


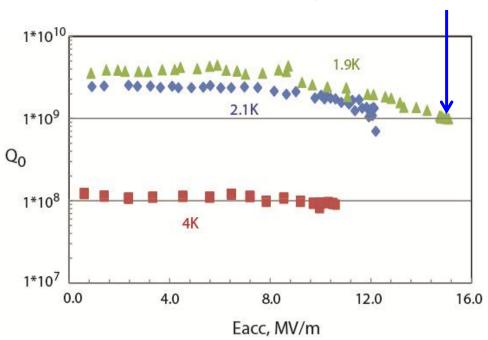


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### 2.1 GHz cavity test result

### Quench at 15 MV/m





$$E_{peak}/E_{acc} = 2.22$$
  
 $B_{peak}/E_{acc} = 8.55 \text{ mT/(MV/m)}$ 

Max.  $B_{peak} \sim 130 \text{ mT}$ 

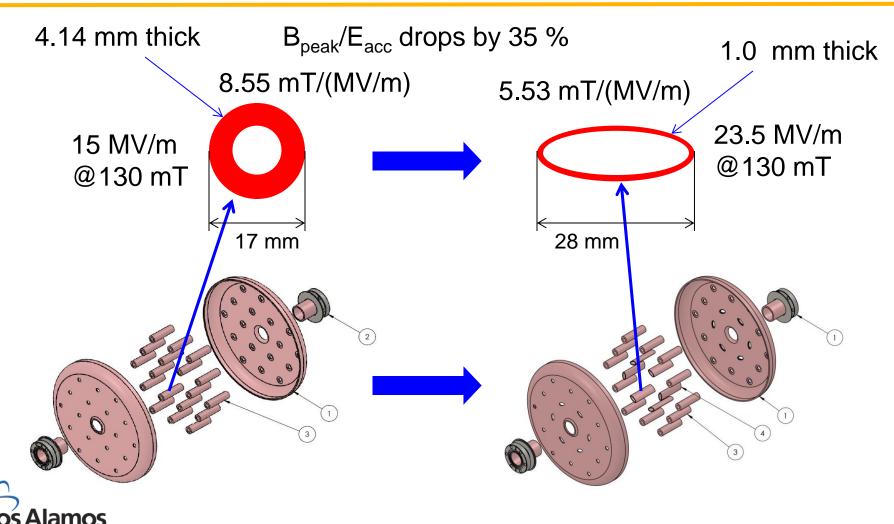


PRL 109 (2012) 164801

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# Plan to increase the accelerating gradient by changing the cross section of the first row pipes



### **Acknowledgements**

#### LANL

- Mike Borden, Jim O'Hara and other AOT-MDE group members especially vacuum team members for their support and encouragement
- Alan Shapiro (retired) and Mike Madrid (retired) for all their work at the SRF Lab.
- Roland Schulze for AES/XPS analyses
- Marilyn Hawley for AFM analyses
- Nestor Haberkorn for magnetization measurements (currently with CNEA-Centro Atómico Bariloche, Argentina)
- Yingying Zhang and Quanxi Jia for Polymer Assisted Deposition (PAD) studies on NbN and MoN. Dr. Zhang is currently with Tsinghua University, Beijing, China.
- Leonardo Civale for leading the magnetization measurements and many useful discussions
- Dave Devlin for setting up the MgB<sub>2</sub> costing system and other coating tests
- Alberto Canabal (now with TriQuint Semiconductor) and Grigory Eremeev (now with JLAB) for testing samples at SLAC and cavities at LANL
- Alp Findikoglu for measuring R<sub>s</sub> of MgB<sub>2</sub> samples using parallel plate technique.



### Acknowledgements (cont.)

#### ANL

 Mike Pellin and Thomas Proslier for ALD coating of alumina and other studies on dielectric materials

#### SLAC

 Sami Tantawi, Jiquan Quo (now at JLAB), Valery Dolgashev, Dave Martin and Charlie Yoneda for preparation and testing samples

#### Jlab

Binping Xiao for measuring RF surface impedance of MgB<sub>2</sub> samples using their 7.5 GHz SIC system. Also, Anne-Marie Valente-Feliciano for preparing ECR Nb films.

### Cornell University

- Alexander Romanenko for conducting TE<sub>01</sub> mode cavity measurements to evaluate R<sub>s</sub> vs. H for MgB<sub>2</sub> samples. Dr. Romanenko is currently with FNAL.
- Hasan Padamsee for useful discussions and encouragement.

#### Temple University

Xiaoxing Xi and his group members for preparing MgB<sub>2</sub> samples with HPCVD.

### **Acknowledgements (cont.)**

#### KEK, Tsukuba, Japan

- Akira Yamamoto, Kenji Ueno, Masashi Yamanaka and Seiya Yamaguchi for support while at Tajima was at KEK on leave from LANL. Hitoshi Inoue for hydroforming and fabricating copper cavities.
- Shigeki Kato for coordination with a company in Japan for mechanical and chemical polishing of copper cavity

### NIMS, Tsukuba, Japan

- Akiyoshi Matsumoto for coordination with Kagoshima U. for MgB<sub>2</sub> coating and other support.
- Eiichiro Watanabe, Hirotaka Ohsato and Daiju Tsuya for ALD coating of alumina
- Minoru Tachiki and Hideki Abe for various discussions while Tajima was at NIMS in 2000.

### Kagoshima University, Japan

 Toshiya Doi and his students (Takafumi Nishikawa, Tomoaki Nagamine and Kazuki Yoshihara) for coating MgB<sub>2</sub> films using E-beam co-evaporation (Dr. Doi is currently with Kyoto University)

### **Acknowledgements (cont.)**

#### University of Wollongong, Australia

Yue Zhao for coating MgB<sub>2</sub> using PLD technique

### Superconductor Technologies, Inc. (STI)

 Brian Moeckly and Chris Yung for coating MgB<sub>2</sub> using reactive co-evaporation technique. (Dr. Moeckly is currently not with STI.)

### Institute for Super Materials, ULVAC, Japan

 Tomohiro Nagata for SIMS analysis for CED Nb films and useful discussions (Dr. Nagata is currently with KEK on leave from ULVAC)

### Alameda Applied Sciences Corporation, San Leandro, CA

Mahadevan Krishnan and Colt James for coating copper cavities with CED technique

#### External sponsors

- DOE/NP 2010 present
- DTRA 2008-2010

